



# A CONSTANT VOLUMETRIC-FAILURE-STRAIN EROSION FOR DETERMINING THE EFFECT OF INERTIA AND STRAIN RATE ON THE CRUSHING STRENGTH OF A CELLULAR CONCRETE

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**Abstract:** The effect of inertia and strain rate on the failure of a cellular autoclaved aerated concrete ( $600 \text{ kg m}^{-3}$ ) was investigated using MAT\_096 material model together with a constant volumetric-failure-strain erosion criterion in the LSDYNA. A rate insensitive, constant compressive yield stress, and a rate sensitive, variable compressive yield stress, model were implemented and the results of models were compared with those of experimental compression tests conducted at similar strain rates, between  $2 \times 10^{-3} \text{ s}^{-1}$  and  $\sim 4150 \text{ s}^{-1}$ . Results have shown an “s-type” compressive strength relation with strain rate, broadly composing of three distinct regions: a lower-velocity-dependent-strength region at the quasi-static velocities, a higher-velocity-dependent-strength region at intermediate velocities and again a lower-velocity-dependent-strength region above  $\sim 1150 \text{ s}^{-1}$ . In experimentally tested samples, a shock fracture strength was presumed to be reached in the higher velocity-dependent strength region, resulting in a cut-off DIF value (2.78), while in numerically tested samples, the compressive strength increased with increasing strain rate in the third region. One dimensional state of strain condition above a critical velocity was also shown numerically. The stress triaxiality increased to 0.66 between 1 and  $30 \text{ m s}^{-1}$ , reaching a fully constraint 1D state of strain condition above  $30 \text{ m s}^{-1}$ . In accord with this, the numerical failure mode, as with that of experiments, switched from an axial- to a radial-dominated cracking after  $\sim 20 \text{ m s}^{-1}$ . Finally, the strain rate dependent compressive strength was numerically shown as partly arising due to the change of deformation state from a 1D state of stress to a 1D state of strain and partly due to the intrinsic rate sensitivity of cellular concrete.

**Keywords:** *Autoclaved aerated concrete, modelling, compressive strength, inertia, strain rate*

**Introduction.** There have been numerous experimental and numerical investigations on the strain rate dependent compressive strength of concrete. A summary on the strain rate dependent compressive strength of concrete can be found in a recent review article [1]. Briefly, the dynamic increase factor (DIF=dynamic facture strength/static fracture strength) of concrete varies between 1 and 2.5, from static to dynamic strain rates, with a sudden increase after about  $100 \text{ s}^{-1}$  [1]. A relatively low dependence of DIF on strain rate until about  $100 \text{ s}^{-1}$  was ascribed to both the strain-rate dependent growth of tensile micro cracks, known as thermally activated facture mechanism, and the viscous behavior of the bulk material between cracks known as Stefan effect [2-4]. The thermally activated facture mechanism is explained as follows. The energy needed for crack opening is much higher than the energy needed for crack growth at quasi-static strain rates. While, there is less time at high strain rates for both crack opening and growth, causing an increase in fracture strength and the number of micro cracks formed as compared with quasi-static strain rates. It was argued that inertial effects become predominant at the strain rates higher than  $10 \text{ s}^{-1}$  [5]. At increasing strain rates, an elastically deforming structure cannot expand in transverse direction (Poisson’s expansion) due to radial inertia restraint. Radial inertia imposes a confinement pressure on deforming structure, transforming the deformation from a uniaxial state of stress to a uniaxial state of strain. The rapid increase of the strength of concrete after about a critical strain rate is therefore ascribed to the development of a

uniaxial state of strain [6-10]. The variations in the DIF values of different studies performed at similar strain rates were further attributed to the variations in the extent of radial and axial inertia between the concrete samples tested [11, 12]. The reported data on the compressive strength of concrete included the forces due to both axial and radial inertia.

An “s-type” dependence of the fracture strength of brittle materials on strain rate was reported [13]. On an “s-type” curve, there are two turning points: (1) from a low-strain rate-dependent strength region to a high-strain rate-dependent strength region and (2) from a high-strain rate-dependent strength region to again a low-strain rate-dependent strength region. The strength of limestone increased slowly with increasing strain rate up to  $10^3 \text{ s}^{-1}$ ; thereafter, increased sharply, approaching the shock fracture strength [6]. The strain rate dependent fracture strength well above  $10^3 \text{ s}^{-1}$  was proposed to resemble the strain dependent fracture strength below  $10^3 \text{ s}^{-1}$ , while the rapid increase of fracture strength around  $10^3 \text{ s}^{-1}$  was ascribed to the transformation of deformation from a uniaxial state of stress to a uniaxial state of strain. The first and second turning points were reported sequentially  $10^2$  and  $10^4 \text{ s}^{-1}$  for concrete [14]. Also, in accord with above, the current concrete models have recently adapted a cut-off value of 2.94 to cap DIF above  $300 \text{ s}^{-1}$  [15]. But, it is not clear whether or not this capping occurs in or after the high-strain rate-dependent strength region. Alternatively, it is proposed that the capping may occur because concrete reaches its ultimate dynamic strength before the second turning point in the high-strain rate-dependent strength region [4].

The aim of this study was to experimentally and numerically investigate the strain rate sensitive compressive strength of an aerated autoclaved concrete (AAC) using a constant volumetric-failure-strain erosion criterion in the MAT\_096 material model of the LSDYNA. Once an element reached a critical volumetric-failure-strain corresponding to that of quasi-static strain rate it was eroded. The used erosion criterion and the associated material model were relatively simple, requiring few experimental input parameter and noted to predict the trends of experimental stress-strain curves with strain rate. The model results were further verified with the compression test results between quasi-static ( $2 \times 10^{-3} \text{ s}^{-1}$ ) and dynamic ( $\sim 4150 \text{ s}^{-1}$ ) strain rates. Modified Split Hopkinson Pressure Bar (SHPB) tests, so called the direct impact tests, were performed to achieve strain rates above  $1000 \text{ s}^{-1}$ . Two different modelling approach called Model 1 and Model 2 were investigated. In the first, a strain rate independent material model was implemented to determine the effect of axial and radial inertia on the fracture strength merely. In the second, a strain rate dependent compressive strength was used as an input to the material model to show the effect of strain rate. Since the investigated upper dynamic strain rates were higher than those of the second turning point of concrete on an “s-type” curve, the model and experimental results allowed the analysis of the strain rate dependent-fracture strength at above these critical strain rates using both approach. Finally, as there has been, so far, no numerical studies on the strain rate dependent compressive strength of these materials and few experimental investigation on the dynamic response [16-19], the results of present study are expected to contribute to the knowledge on modelling dynamic mechanical response of such brittle cellular materials.

## **Tests and models.**

### ***Tests***

Quasi-static and dynamic compression test samples were  $\sim 19.4 \text{ mm}$  in diameter and  $26 \text{ mm}$  in length. The test samples' diameter was determined by the bar diameter of the used SHPB. Quasi-static compression tests were performed in a Shimadzu AG-X Universal Test machine at  $5 \times 10^{-5}$ ,  $5 \times 10^{-4}$ , and  $5 \times 10^{-3} \text{ m s}^{-1}$ , corresponding to strain rates of  $\sim 2 \times 10^{-3}$ ,  $\sim 2 \times 10^{-2}$  and  $\sim 2 \times 10^{-1} \text{ s}^{-1}$ , respectively. Low-velocity compression tests were performed in a FRACTOVIS drop-weight test device using a flat-ended striker at about  $1 \text{ m s}^{-1}$ , corresponding to a strain rate of  $\sim 35 \text{ s}^{-1}$ . The dynamic compression tests were performed in a SHPB apparatus, having  $19.4 \text{ mm}$ -diameter Inconel 718 incident ( $3110 \text{ mm}$ ) and transmitter ( $2050 \text{ mm}$ ) bars. Conventional SHPB compression tests were performed at  $8 \text{ m s}^{-1}$  corresponding to  $\sim 185 \text{ s}^{-1}$ . In the direct impact tests, the striker bar (Inconel bar  $500 \text{ mm}$ -long and

aluminum bar 200 mm-long) directly impinged on the test sample inserted to the end of the SHPB incident bar. The direct impact SHPB tests were performed at 10, 30 and 108 m s<sup>-1</sup> corresponding to ~385, ~1150 and ~4150 s<sup>-1</sup>, respectively. At least 8 samples were tested at each strain rate. The details of the direct impact tests and the used SHPB test device are given elsewhere [20-22]. The tensile strength of AAC sample was determined by the indirect tensile Brazilian tests. In these tests, the cylindrical compression test samples (3 tests), 19.4 mm in diameter and 26 mm in length, were compressed laterally in a Shimadzu AG-X Universal Test Machine. The tensile strength ( $\sigma_t$ ) was then calculated as

$$\sigma_t(t) = \frac{2P}{\pi DL} \quad (1)$$

where  $P$ ,  $D$  and  $L$  are sequentially the fracture load and the diameter and thickness of sample.

### **Models**

The quasi-static and high strain rate compression and direct impact tests were simulated in the non-linear explicit finite element code of LS-DYNA. The quasi-static model was briefly composed of top and bottom compression test platen (tool steel) and sample as seen in Figure 1(a). Each compression platen was modelled using 6 mm-long and 2 mm-wide 19200 solid elements and MAT020\_RIGID material model ( $E=210$  GPa and  $\nu=0.3$ ). The rotations and the translations of compression test platens were restricted in all directions, except the axial translation of the top platen in z-direction was kept constant by PRESCRIBED\_MOTION\_RIGID card at  $5 \times 10^{-3}$  m s<sup>-1</sup>, the same as the quasi-static tests. The contacts between compression test platens and sample were defined by AUTOMATIC\_SURFACE\_TO\_SURFACE. The mass scaling was implemented in the quasi-static simulation by using CONTROL\_TIMESTEP card. The model was initially simulated without mass scaling in order to determine time-step. The determined time-step was then multiplied by a factor until kinetic energy became much smaller than internal energy. A mass scaling factor of 1000 was determined by following above procedure. In the SHPB test model, the striker, incident and transmitter bars were modelled using 15x2 mm 7000, 28980 and 19180 solid elements, respectively (Figure 1(b)). The striker bar velocities in the SHPB model were 1 and 8 m s<sup>-1</sup>. The drop-weight test at ~1 m s<sup>-1</sup> was also modelled with the SHPB compression test model. The lengths of Inconel 718 striker, incident and transmitter bar were sequentially 500 mm, 3110 mm and 2050 mm, the same as the used SHPB. The contacts between the bars and sample and between the striker and incident bar were defined by AUTOMATIC\_SURFACE\_TO\_SURFACE and AUTOMATIC\_SINGLE\_SURFACE, respectively. The impact velocities in the SHPB direct impact simulations were 10, 20, 30, 60 and 108 m s<sup>-1</sup>. The models at 10 and 20 m s<sup>-1</sup> were implemented with 500 mm-long Inconel 718 striker bar, while the tests at 30, 60 and 108 m s<sup>-1</sup> were implemented using 200 mm-long aluminum striker bar, the same as the tests. The incident and striker bars in these tests were again modeled using 15x2 mm solid elements, whereas aluminum striker bar was modelled using 5x2 mm 10080 solid elements (Figure 1(c)). The contact between the sample and incident bar was defined by AUTOMATIC\_SURFACE\_TO\_SURFACE and the contact between the striker and incident bar was defined by AUTOMATIC\_SINGLE\_SURFACE. The velocity of striker bar was defined by the VELOCITY\_GENERATION card in LS-DYNA. Inconel 718 striker, incident and transmitter bar and aluminum striker bar were simulated using MAT001\_ELASTIC material model (Inconel 718:  $E=207$  GPa,  $\nu=0.33$  and  $\rho=7850$  kg m<sup>-3</sup> and aluminum:  $E=71.7$  GPa,  $\nu=0.33$  and  $\rho=2810$  kg m<sup>-3</sup>). The static and dynamic friction coefficients were taken 0.2 and 0.1 for lubricated surfaces and 0.3 and 0.2 for non-lubricated surfaces, respectively.

The cylindrical AAC sample was modelled using 38400 solid elements and MAT096\_BRITTLE\_DAMAGE material model. The used material model was also previously applied to simulate the failure of concrete [23]. The material model allowed to admit progressive degradation of tensile and shear strengths across smeared cracks initiated under tensile loads [24]. The compressive failure was governed by a simplistic J2 flow correction [25]. The damage occurred was handled by treating 4-rank elastic stiffness tensor as an evolving internal variable. The main material model parameters are the Elastic modulus, Poisson's ratio, the initial principal tensile strength ( $f_n$ ), the initial shear traction ( $f_s$ ), the fracture toughness of the material ( $g_c$ ), shear retention factor ( $\beta$ ), the viscosity of the material ( $\eta$ ) and uniaxial compressive yield stress ( $\sigma_y$ ) [26]. ERODING\_SINGLE\_SURFACE was applied to AAC sample by MAT000\_ADD\_EROSION parameter in both SHPB and direct impact test models. The volumetric-failure-strain (corresponding to compressive strength) at a quasi-static strain rate (0.0117) was used as the erosion parameter in MAT\_ADD\_EROSION. The volumetric strain ( $\Delta$ ) is

$$\Delta = \varepsilon_x + \varepsilon_y + \varepsilon_z \quad (2)$$

where  $\varepsilon_x$ ,  $\varepsilon_y$  and  $\varepsilon_z$  are the normal strains in x, y and z-axis. The model stresses were determined at the distal-end and impact-end contact areas, at the center and surface elements at the contact areas and on the incident bar at the strain gage location of the test (Figure 1(d)). The determined material parameters of AAC are tabulated in Table 1.

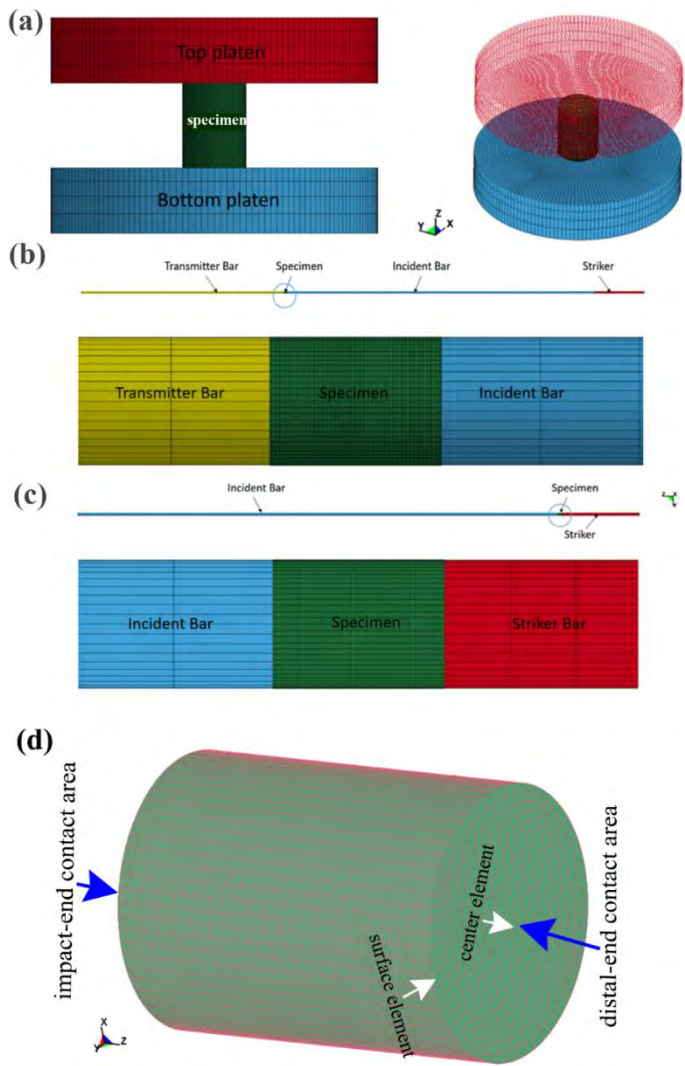


Fig. 1. 2D and 3D models of (a) quasi-static compression, (b) SHPB compression, (c) SHPB direct impact tests and (d) sample with the location of numerical stress measurement

Table 1. Material model parameters of AAC

Parameter	Value
Density, $\rho$ (kg m <sup>-3</sup> )	600
Young's Modulus, $E_B$ (GPa)	0.32
Poisson's Ratio, $\nu$	0.2
Tensile Limit (MPa)	1
Shear Limit (MPa)	2.56
Compressive yield stress (MPa)	5.11
Fracture Toughness (N m <sup>-1</sup> )	3.47
Shear retention factor	0.03

A shear retention factor of 0.03, typical for concrete, was taken from reference [23] and all other material parameters were extracted for the tests at a quasi-static strain rate of  $0.002 \text{ s}^{-1}$ . The viscosity was taken zero in order to exclude the effect of strain rate in Model 1, while the compression yield stress corresponding to each velocity were entered in the material model for the strain rate sensitive model, Model 2.

## Results and Discussion.

### *Quasi-static and high strain rate tests*

Three representative stress-strain curves of quasi-static compression tests at  $5 \times 10^{-5}$  and  $5 \times 10^{-3} \text{ m s}^{-1}$ , drop-weight compression tests at  $1 \text{ m s}^{-1}$  and SHPB direct impact tests at  $8 \text{ m s}^{-1}$  are shown in Figure 2(a). Note that in the SHPB direct impact tests, the stress was measured from a strain gage 1 m away the sample/incident bar contact area. The equilibrium in SHPB test was further checked by using the following relation [27]:  $3t_{tr} = \varepsilon_f l/v$ ; where  $t_{tr}$  and  $\varepsilon_f$  are the transit time ( $l/\sqrt{E/\rho}$ ) and failure strain of the sample and  $v$ ,  $l$ ,  $E$  and  $\rho$  are the impact velocity and the length, elastic modulus and density of sample, respectively. Using  $\varepsilon_f=0.012$ ,  $l=26 \text{ mm}$ ,  $\rho=600 \text{ kg m}^{-3}$  and  $E=0.7 \text{ GPa}$  (Figure 2(a)), one can arrive a critical velocity of  $\sim 5 \text{ m s}^{-1}$  ( $t_{tr}=22 \text{ }\mu\text{s}$ ) above which there will be no stress equilibrium in the SHPB test. In the SHPB compression tests at  $8 \text{ m s}^{-1}$ , an aluminum disc of 1 mm thick and 10 mm in diameter was used as a pulse shaper [9] in the front of the incident bar to attain stress equilibrium. In these tests, the failure time (100  $\mu\text{s}$ ) was more than 4 times the wave transit time, showing nearly a stress equilibrium condition and corresponding to a strain rate of  $185 \text{ s}^{-1}$  at failure strain (Figure 2(b)). While, the tests at  $10 \text{ m s}^{-1}$  as seen in Figure 2(b) ( $\varepsilon_f l/v=45 \text{ }\mu\text{s}$ ) and above are non-equilibrium tests. In these tests, the deformation is represented by  $\frac{v}{l}t$  instead of strain. Figure 2(c) shows representative equilibrium stress- $\frac{v}{l}t$  curves at  $5 \times 10^{-5}$  and  $1 \text{ m s}^{-1}$  and representative non-equilibrium stress-strain curves at 10, 30 and  $108 \text{ m s}^{-1}$ . Non-equilibrium stress- $\frac{v}{l}t$  curves in the same figure are shown for the comparison between compressive stresses. The maximum stresses in the stress-strain curves of Figures 2(a) and in the stress- $\frac{v}{l}t$  curves of Figure 2(c) are taken as the compressive strengths. As noted in the same figures, the compressive strength increases as the velocity increases between  $5 \times 10^{-5}$  to  $30 \text{ m s}^{-1}$ , while, the compressive strengths at 30 and  $108 \text{ m s}^{-1}$  are very much similar. The elastic modulus was determined from the initial slopes of the curves, from  $5 \times 10^{-5}$  to  $8 \text{ m s}^{-1}$ . The elastic modulus increases from  $\sim 0.32 \text{ GPa}$  at quasi-static velocity to  $\sim 0.7 \text{ GPa}$  at  $8 \text{ m s}^{-1}$  (Figure 2(a)). The slopes of stress- $\frac{v}{l}t$  curves of the tests at 30 and  $108 \text{ m s}^{-1}$  are comparatively lower as these are non-equilibrium tests. The variation of compressive failure strain (corresponding to the compressive strength) with velocity until about  $8 \text{ m s}^{-1}$  (SHPB compression equilibrium test) is shown in Figure 2(d). The compressive failure strain starts to decrease at  $8 \text{ m s}^{-1}$  and an average compressive failure strain of 0.017 is taken for the quasi-static tests. Assuming the SHPB test at  $8 \text{ m s}^{-1}$  is confined state (no lateral expansion of sample), the volumetric fracture strain are calculated for the tests below this velocity. As shown in Figure 2(d), the average volumetric failure strain is about 0.0117 and this value was used in the modelling. Three tensile stress-displacement curves of the Brazilian tests are shown in Figure 2(e), together with the picture of an undeformed and a deformed, centrally-fractured test sample. The mean tensile strength was determined  $\sim 1 \text{ MPa}$  at  $2 \times 10^{-3} \text{ s}^{-1}$ . Dynamic tensile fracture tests were also performed in the SHPB at  $10 \text{ m s}^{-1}$  (not shown here) and the mean tensile strength in these tests increased to  $\sim 1.5 \text{ MPa}$ , showing a strain rate dependent splitting behavior of tested AAC samples.

A constant volumetric-failure-strain erosion for determining the effect of inertia and strain rate on the crushing strength of a cellular concrete

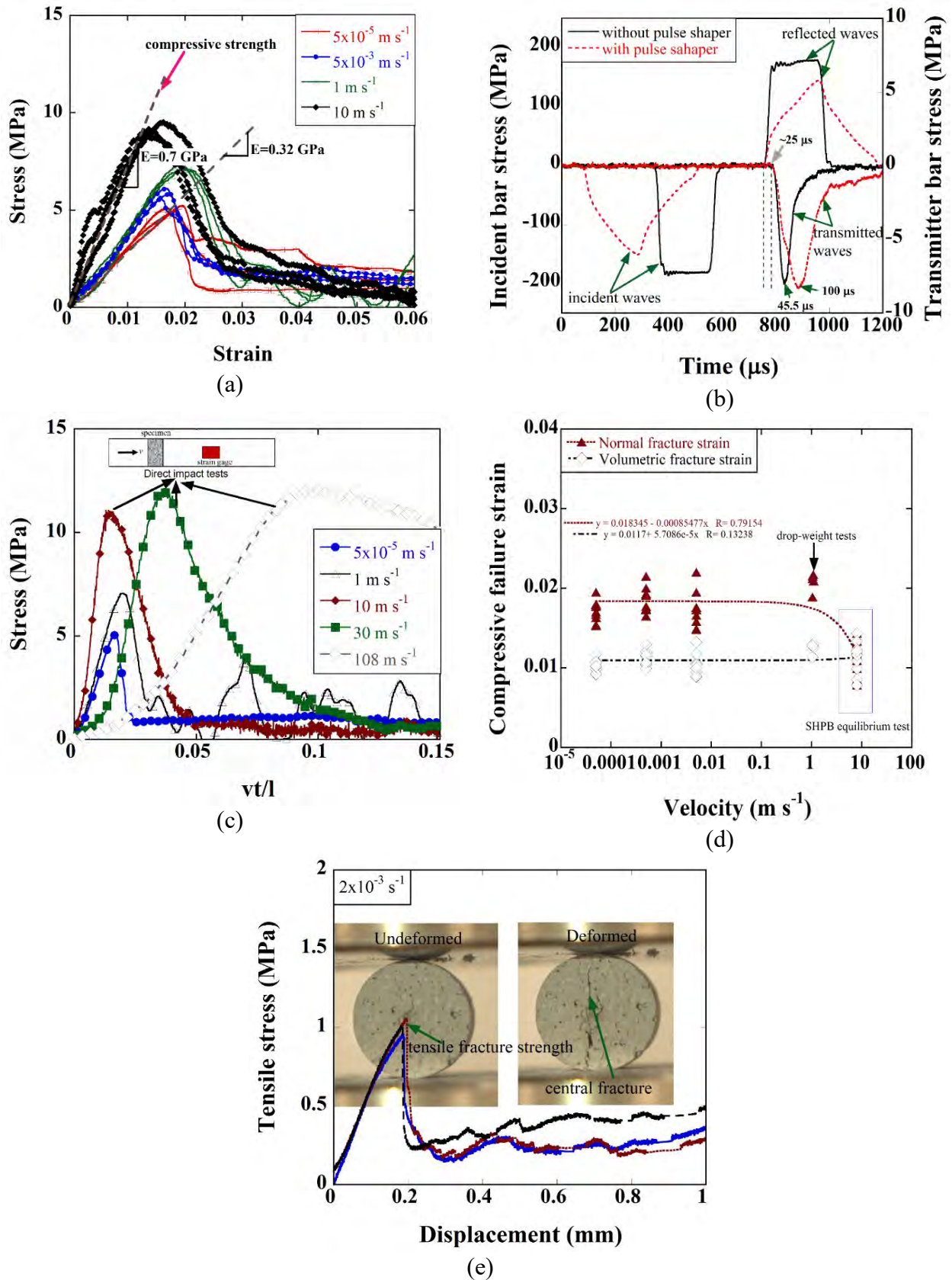


Fig. 2. (a) three stress-strain curves at  $5 \times 10^{-5}$ , 1 and  $10 \text{ m s}^{-1}$ , (b) SHPB tests bar stresses with and without pulse shaper at  $8 \text{ m s}^{-1}$ , (c) typical stress-strain curves at  $5 \times 10^{-5}$ , 1, 8, 30 and  $108 \text{ m s}^{-1}$ , (d) the variation of compressive strain with velocity and (e) the quasi-static tensile stress-displacement curves of three Brazilian tests

The failure at quasi-static velocities occurs by the initiation of a single large axial cracks at the bottom compression test platen as shown by the arrows in Figure 3(a). Additional axial cracks are then formed as the upper compression platen continues to compress the sample until about large displacements.

The samples tested at 1 and 10 m s<sup>-1</sup> also fail similarly, forming few axial cracks starting at the impact-end and/or striker bar/sample contact area, respectively (Figure 3(b) and (c)). However, the extensive cracking of the sample at the impact-end composing of both axial and radial cracks is seen at 30 and 108 m s<sup>-1</sup> (Figures 3(d) and (e)). The number of cracks also significantly increase at these velocities, clearly indicating the effect of inertia on the fracture behavior of the tested AAC.



Fig. 3. The deformation pictures of the samples tested at (a)  $5 \times 10^{-5}$ , (b) 1, (c) 10, (d) 30 and (e) 108 m s<sup>-1</sup>

The variation of the compressive strength with strain rate is shown in Figure 4(a) and may be considered in three sequential distinct regions: a lower velocity-dependent strength region (Region 1) at quasi-static and low velocities, a higher velocity-dependent strength region (Region 2) at intermediate velocities between broadly 10 and 30 m s<sup>-1</sup> and a constant strength region (Region 3), likely above 30 m s<sup>-1</sup>. These regions are shown by the numbered-circles in Figure 4(a). The International Federation for Structural Concrete (CEB) recommended two empirical equations to define the DIF of concrete strength as [28]

$$DIF = \frac{\sigma_d}{\sigma_s} = \left( \frac{\dot{\epsilon}_d}{\dot{\epsilon}_s} \right)^{1.026\alpha} \quad \dot{\epsilon}_d \leq 30 \text{ s}^{-1} \quad (3)$$

$$DIF = \frac{\sigma_d}{\sigma_s} = \gamma \dot{\epsilon}_d^{1/3} \quad \dot{\epsilon}_d > 30 \text{ s}^{-1} \quad (4)$$

where,  $\dot{\epsilon}_d$  and  $\dot{\epsilon}_s$  are the dynamic and static strain rates, respectively. The value of  $\dot{\epsilon}_s$  is  $3 \times 10^{-5} \text{ s}^{-1}$ ,  $\gamma = 10^{6.156\alpha-2}$  and  $= \frac{1}{(5+\frac{\sigma_s}{10})}$ . Fitting the compressive strength values with Eqn. 3, between the lowest quasi-static strain rate and  $35 \text{ s}^{-1}$  ( $1 \text{ m s}^{-1}$ ), and with Eqn. 4 between  $35 \text{ s}^{-1}$  ( $1 \text{ m s}^{-1}$ ) and  $1150 \text{ s}^{-1}$  ( $30 \text{ m s}^{-1}$ ) yield a fracture strength of 4.225 MPa at the reference strain rate, 0.381 for the value of  $\gamma$  and 0.037 for the value of  $\alpha$  as depicted in Figure 4(a). As shown in the same figure, the critical strain rate is predicted broadly  $100 \text{ s}^{-1}$  ( $\sim 5 \text{ m s}^{-1}$ ) for rapidly increased compressive strength and  $380 \text{ s}^{-1}$  ( $\sim 18 \text{ m s}^{-1}$ ) or the constant stress region by using Eqn. 4. The compressive strength in Region 2 is also fitted with the following more general equation:  $\frac{\sigma_d}{\sigma_s} = \beta \dot{\epsilon}_d^n$ , where  $n=1.113$  and  $\beta=0.134$ , resulting in a critical strain rate of  $\sim 20 \text{ s}^{-1}$  ( $\sim 1 \text{ m s}^{-1}$ ) for increased compressive strength in Region 2 and  $\sim 760 \text{ s}^{-1}$  ( $\sim 30 \text{ m s}^{-1}$ ) for the constant stress Region 3. This power equation is noted to be better fitted in Region 2 with the mean DIF values than Eqn. 4 proposed by the CEB as shown in Figure 4(b). A cut-off DIF value of 2.78 above  $\sim 1000 \text{ s}^{-1}$  ( $18\text{-}30 \text{ m s}^{-1}$ ) corresponding to a compressive strength of  $\sim 11.5 \text{ MPa}$ , is also determined for the tested AAC. Figure 4(c) shows the determined compressive strength values of the tested AAC sample using Eqn. 3 at quasi-static velocities and 1, 10, 20, 30, 60 and 108 m s<sup>-1</sup>. The strength values corresponded to specific test velocities by excluding strength enhancement in Region 2 were used as compressive strength input to the model (Model 2) to determine the strain rate effect on the compressive strength. The experimental mean compressive

strength values are further tabulated in Table 2 together with the corresponding velocity and strain rate.

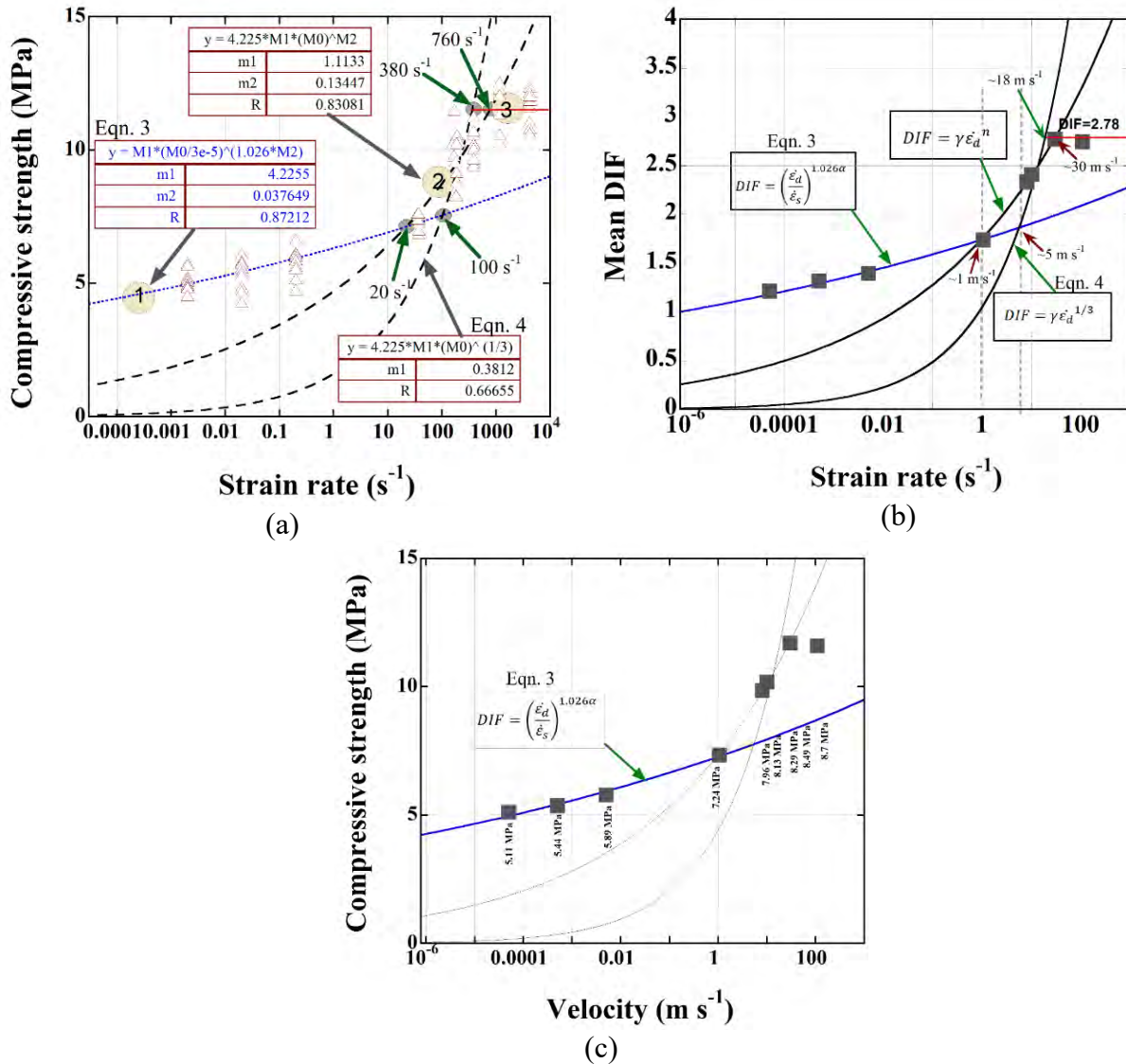


Fig. 4. Experimental (a) compressive strength versus strain rate and fitting with Eqns. 1 and 3, (b) mean DIFs versus velocity and (c) the predicted compressive yield stress of Model 2 at various velocities

### Numerical quasi-static and high strain rate compression tests

Figure 5(a) shows the Model 1 numerical stress- $\frac{v}{l}t$  curves at different velocities using a constant compressive strength material model (5.11 MPa) determined at the quasi-static strain rate of  $5 \times 10^{-5} \text{ m s}^{-1}$ . Note that only quasi-static compression test at  $0.005 \text{ m s}^{-1}$  was quasi-statically modelled and the stress in the direct impact SHPB tests were measured both numerically and experimentally 1 m away the sample/incident bar contact area. As similar with experiments, the numerical compressive strength increases with increasing velocity until about  $30 \text{ m s}^{-1}$ ; then it increases slightly with increasing velocity above  $30 \text{ m s}^{-1}$ . The reduction of the slopes of stress- $\frac{v}{l}t$  curves after about  $10 \text{ m s}^{-1}$  is also seen numerical stress- $\frac{v}{l}t$  curves but the extent of reduction declines as compared with the tests. The strain gage read numerical compressive strength values are 6, 6.11, 7.2, 8.64, 8.82 and 9.05 MPa sequentially at 1, 10, 20, 30, 60 and  $108 \text{ m s}^{-1}$ . Compare to experimental



strengths tabulated in Table 2, Model 1 numerical compressive strength values are significantly lower at the same velocities.

Table 2. The mean fracture strength of the quasi-static, drop-weight, SHPB and direct impact compression tests

Velocity (m s <sup>-1</sup> )	Approximate strain rate (s <sup>-1</sup> )	Test	$\sigma_m$ (MPa)
5x10 <sup>-5</sup>	2x10 <sup>-3</sup>	Quasi-static compression	5.11
5x10 <sup>-4</sup>	2x10 <sup>-2</sup>	Quasi-static compression	5.37
5x10 <sup>-3</sup>	2x10 <sup>-1</sup>	Quasi-static compression	5.89
1	35	Drop-weight	7.34
8	185	SHPB	9.9
10	385	Direct impact	10.2
30	1150	Direct impact	11.70
108	4150	Direct impact	11.60

When the compression strength is taken strain rate dependent that is the corresponding to compressive strength values at each velocity are input to the material model (Model 2), the numerical compressive strengths increase as seen in Figure 5(b).

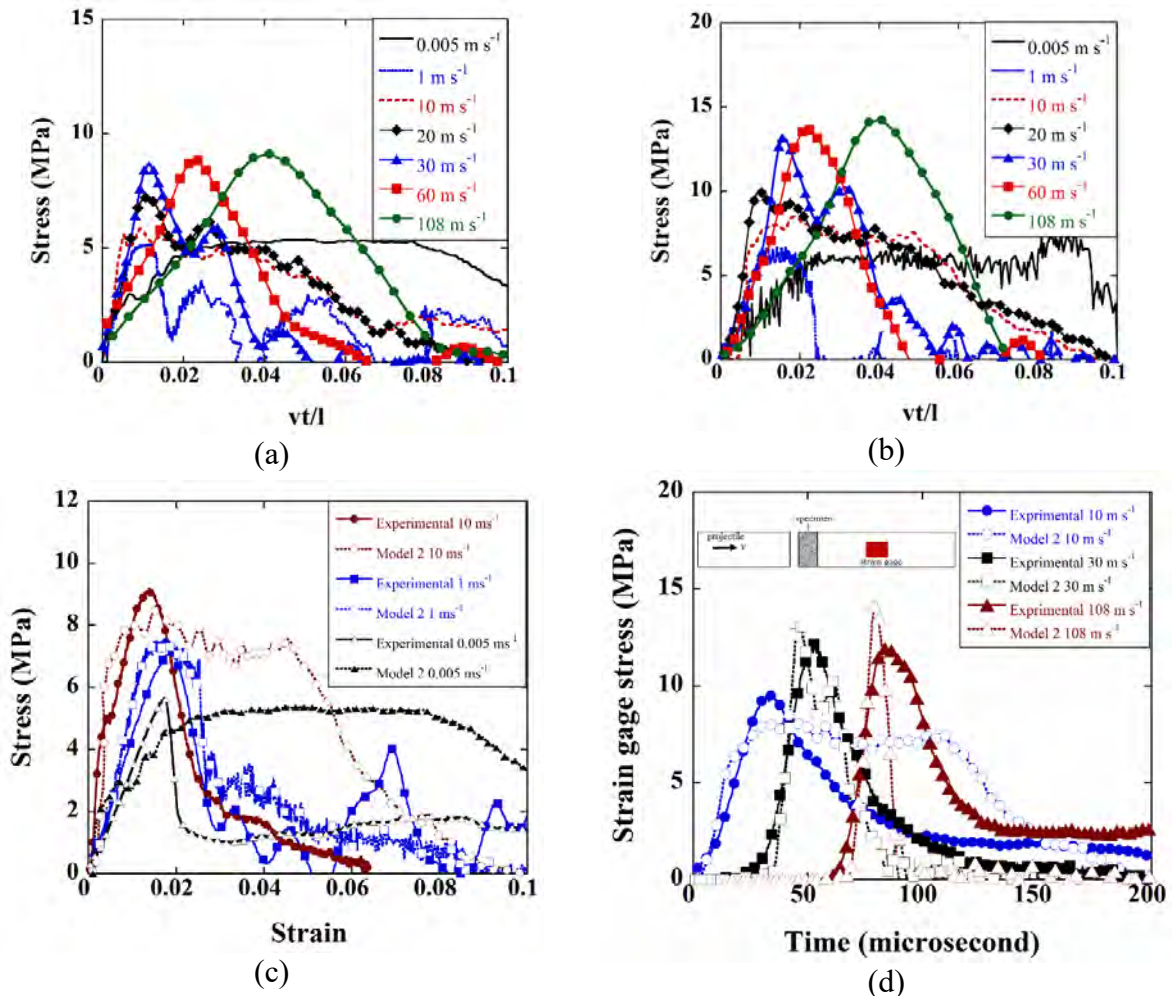


Fig. 5. The numerical stress-strain curves at different velocities; (a) Model 1 and (b) Model 2 and comparison of experimental and model 2 (c) stress-strain and (d) strain gage stress-time curves

Figures 5(c) and 5(d) show the stress-strain and stress-time curves between 0.005 and 8 m s<sup>-1</sup> and between 10 and 108 m s<sup>-1</sup>, respectively. The strain gage read numerical compressive strength values in Model 2 increase to 6.4, 6.7, 8.48, 10, 13.1, 13.68 and 14.2 MPa sequentially at 0.005, 1, 10, 20, 30, 60 and 108 m s<sup>-1</sup> and become much more comparable with the experimental compressive strength values tabulated in Table 2. Also noted in Figures 5(c) and (d), the experimentally measured gage compressive strength values are higher than those of Model 2 until about 10 m s<sup>-1</sup>; thereafter, Model 2 strength values becomes higher at 30 and 108 m s<sup>-1</sup>. The post-failure regions of experimental and model curves are also different, Model 2 show a more progressive failure than the experiments. Despite these differences between the gage-measured stress of models and experiments, Model 2 and Model 1 show pretty well the trends of compressive strength variation with velocity. After this verification, the model compressive strength values were determined at the distal-end and impact-end contact areas as well as the elements at the center and surface of the sample at both distal-end and impact-end.

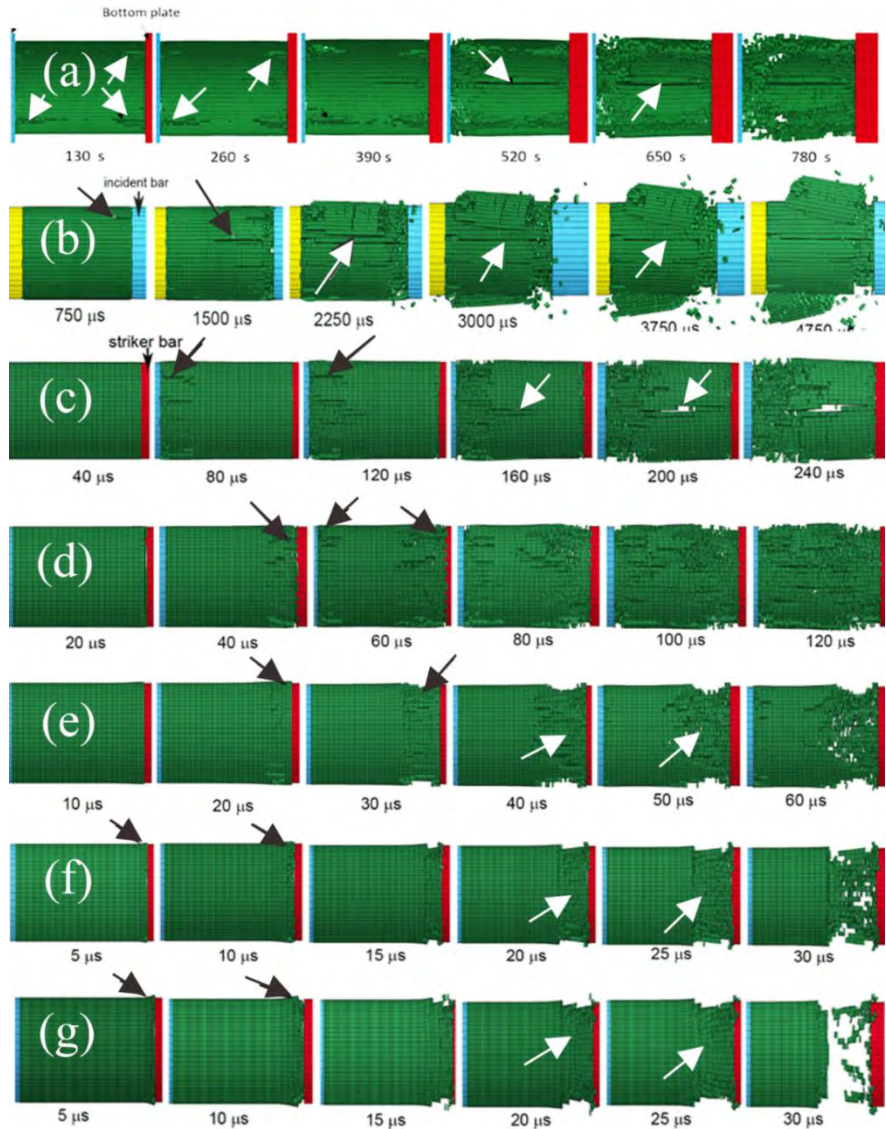


Fig. 6. The numerical deformation pictures of the samples tested at (a) quasi-static, (b) 1, (c) 10, (d) 20, (e) 30, (f) 60, and (g) 108 m s<sup>-1</sup>

Figures 6(a-g) show Model 1 numerical deformation pictures (at various deformation times) at 0.005, 1, 10, 20, 30, 60 and 108 m s<sup>-1</sup>, respectively.

The corresponding 3D views of the characteristic damage progression modes (after about compressive strength) between 1 and 108 m s<sup>-1</sup> are also depicted in Figure 7. As marked by arrows in Figure 6(a-c), the cracks initiate either at one end or at both ends of the samples, while later these initial cracks turn to one or two large axial cracks at increasing times at 0.005, 1 and 10 m s<sup>-1</sup>. This explained numerical axial cracking behavior also seen in Figure 7 at 1 and 10 m s<sup>-1</sup>, and is pretty much comparable with the experimental fracture behavior at the same velocities, as depicted in Figure 3. However, the cracks initiate at the impact-end and are proceeded by radial cracking at 20 m s<sup>-1</sup> and above as shown in Figures 6(d-g) and Figure 7. Again, the numerical failure shows well matching with the experimental failure at 30 and 108 m s<sup>-1</sup> shown in Figure 3. Model 2 deformation and failure

modes at varying velocities were also found very similar with Model 1, except the failure occurred at higher stresses.

Figure 8 shows the variation of numerical distal-end DIFs with the logarithm of velocity. In the same figure, the experimentally measured strain-gage DIFs are also shown for comparison. A rapid increase in the experimental and numerical DIFs after about  $1 \text{ m s}^{-1}$  is clearly seen in the same figure. The increase in the DIFs is noted to continue until about  $30 \text{ m s}^{-1}$ . The experimental DIFs however show a sudden cut-off at about  $30 \text{ m s}^{-1}$ , while both Model 1 and Model 2 DIFs increase with increasing velocity above  $30 \text{ m s}^{-1}$ . The DIFs of Model 1 however increases relatively slowly above  $30 \text{ m s}^{-1}$ , while the increase in Model 2 is comparably higher. This difference is attributed to the velocity dependent compressive yield stress used in Model 2. As Model 1 uses a constant compressive yield stress it shows merely the effect of axial and radial inertia on the DIF values, while Model 2 shows both the effects of inertia and strain rate. The modeling results tend to conclude that inertia is effective with increasing velocity between 1 and  $30 \text{ m s}^{-1}$ . Experimentally, it is presumed that AAC sample reaches a shock compressive strength in Region 2 without entering Region 3. A similar conclusion may be made for Model 1. While, in Model 2, the compressive yield stress also increases in Region 3. Although Model 1 resembles the experimental DIFs by a constant compressive strength in Region 3, Model 2 resembles the experimental DIFs in both Region 1 and Region 2. Nevertheless, both models show an “s-type” DIF-velocity graph, proving the transformation of the deformation state from 1D stress to 1D strain. Lastly, although the proposed power law equation shows well matching with the experimental and model DIFs at lower velocities in Region 2, Eqn. 4 better predicts Model 2 DIFs at higher velocities,  $\sim 30 \text{ m s}^{-1}$ .

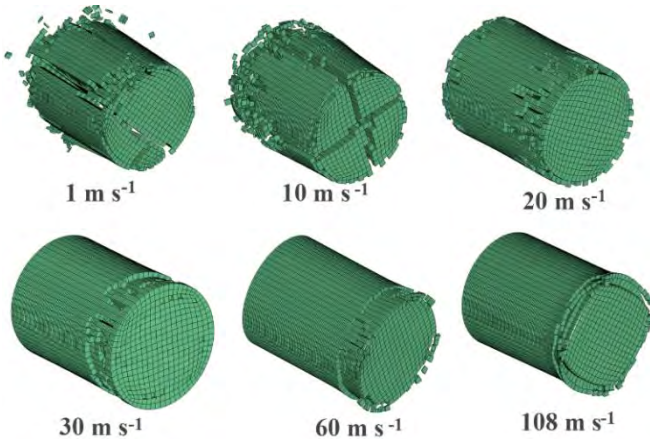


Fig. 7. Model 1 numerical failure modes at different velocities

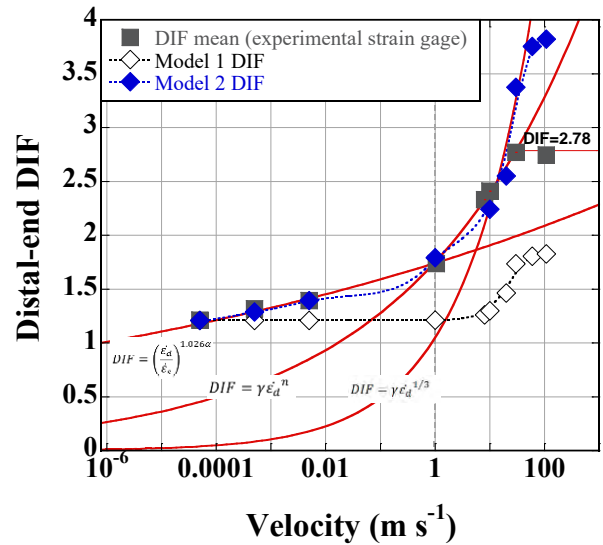


Fig. 8. The variations of experimental and numerical DIFs and numerical contact stresses with velocity

Figures 9 (a-e) show the variation of Model 1 impact-end, distal-end and distal–end center and surface element stresses with time sequentially at  $1, 10, 20, 30$  and  $60 \text{ m s}^{-1}$ . The element stresses were determined from the selected center and surface element at the distal-end, while impact-end and distal-end stresses were calculated from all elements at the surface of the sample. The time difference between the impact and distal-end stresses in the same figures is due to the wave-transit-time of AAC sample. Figures 9(a-e) clearly indicate that the impact-end and distal-end stresses are similar until about  $20 \text{ m s}^{-1}$ , while the center and surface element stresses starts to differentiate at about  $10 \text{ m s}^{-1}$ . The center element stress is higher than the surface element stress until about  $20 \text{ m s}^{-1}$ , then the surface element stress increases over that of the center element stress at higher velocities, proving the transformation of the failure from an axial to a radial fracture.

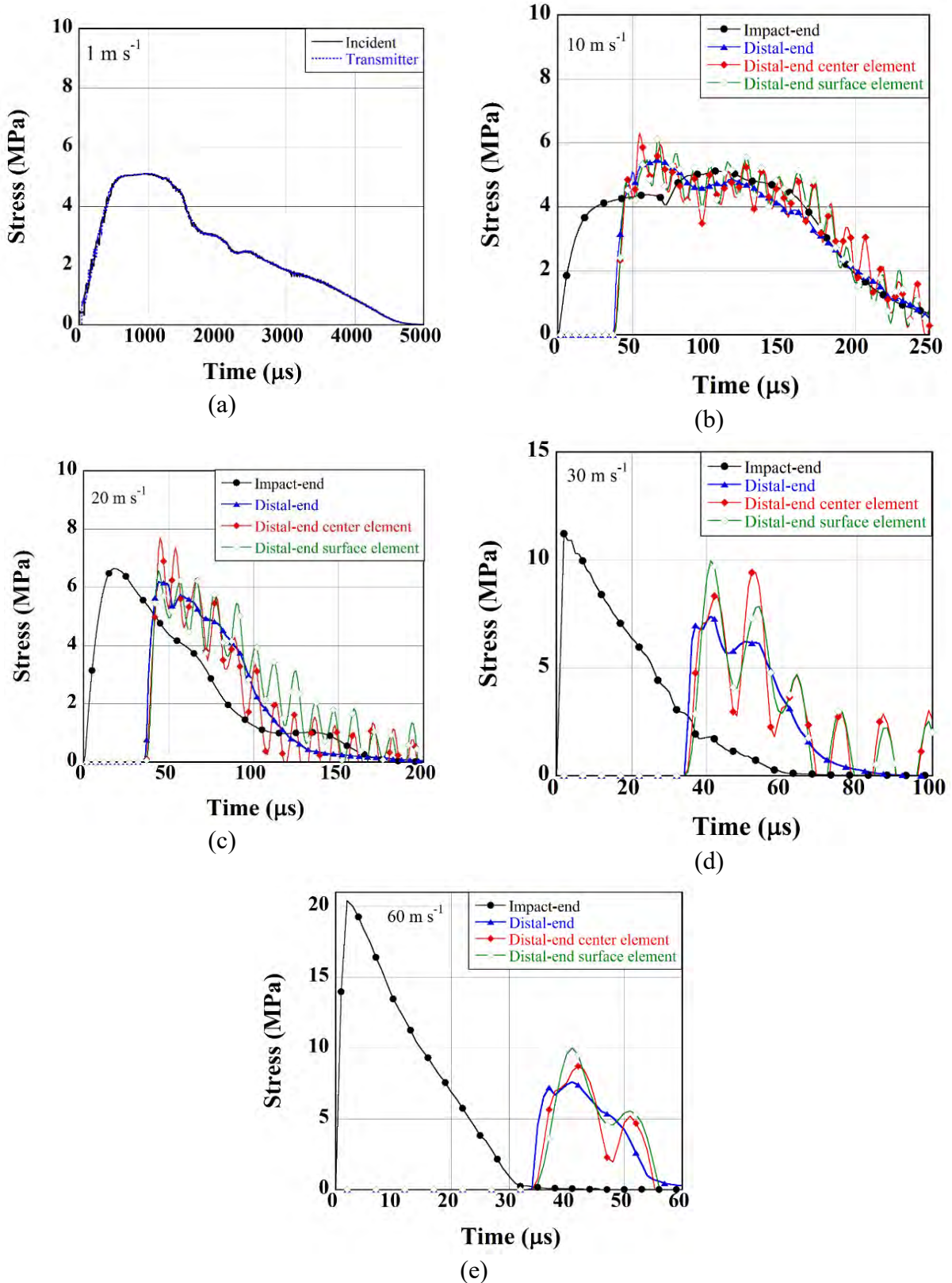


Fig. 9. The variation of Model 1 impact-end, distal-end, distal-end center and surface element and strain gage location stresses with time at (a) 1, (b) 10, (c) 20, (d) 30 and (e) 60 m s<sup>-1</sup>

The stress triaxiality is defined as

$$\eta_{1D \text{ stress}} = \frac{\sigma_x + \sigma_y + \sigma_z}{3\sigma_{eq}} \quad (5)$$

where  $\sigma_x$ ,  $\sigma_y$  and  $\sigma_z$  are stresses on x, y and z directions and  $\sigma_{eq}$  is equivalent stress. The equivalent stress is

$$\sigma_{eq} = \frac{1}{\sqrt{2}} \left[ (\sigma_x - \sigma_y)^2 + (\sigma_y - \sigma_z)^2 + (\sigma_z - \sigma_x)^2 + 6(\tau_{xy}^2 + \tau_{yz}^2 + \tau_{xz}^2) \right]^{\frac{1}{2}} \quad (6)$$

Since  $\sigma_{eq} = \sigma_x$  in the quasi-static test,  $\eta_{1D \text{ stress}} \approx 0.33$ . Assuming fully confined elastic state at high strain rates, the stress triaxiality is

$$\eta = \frac{1+\nu}{3(1-2\nu)} \quad (7)$$

By taking the Poisson's ratio equals to 0.2, the stress triaxiality approaches to  $\sim 0.66$  for a full 1D state of strain. Figure 10 shows the Model 1 and Model 2 distal-end and impact end DIFs together with the Model 1 impact-end surface element stress triaxiality (non-eroded element). The numerical stress triaxiality increases with increasing velocity after about  $10 \text{ m s}^{-1}$  ( $35 \text{ s}^{-1}$ ) and reaches a steady value of 0.66 at about  $30 \text{ m s}^{-1}$  (marked by an arrow in Figure 10). The result clearly indicates a full uniaxial state of strain attainment in the numerically tested samples after about  $30 \text{ m s}^{-1}$  ( $1000 \text{ s}^{-1}$ ). Modelling SHPB tests on a concrete using a pressure dependent strength model previously showed a stress triaxiality ( $\eta$ ) was near 1D state of stress ( $\eta = -0.33$ ) at  $47 \text{ s}^{-1}$ , while it reached 1D state of strain ( $\eta = -0.66$ ) at  $\sim 795 \text{ s}^{-1}$  when  $\nu = 0.2$  [15]. Above this critical strain rate, the sample deformation was completely 1D state of strain. Furthermore, the critical strain rate for the passage to 1D state of strain was shown to depend on the diameter of sample, larger diameter samples showed larger inertial effects hence lower critical strain rate for the complete 1D state of strain [15]. In the present study relatively small samples were used, 19.4 mm in diameter and 26 mm in length. The transformation strain rate from the uniaxial state of stress to the uniaxial state of strain is found  $1150 \text{ s}^{-1}$ , which is also very much consistent with the previous numerical simulations on concrete [15].

Assuming cellular concrete deforms in 1D state of strain by forming a shock front depicted in Fig. 11, the impact end-stress ( $\sigma^*$ ) is governed by the following rigid-perfectly-plastic-locking (r-p-l) model, based on mass and momentum conservation as [29],

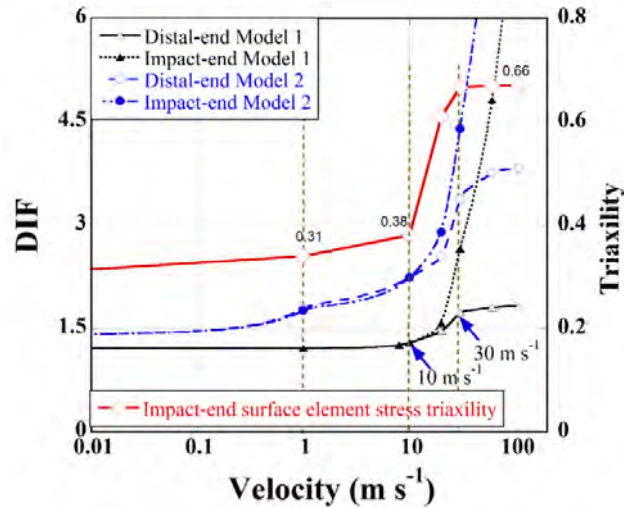


Fig. 10. Experimental and numerical DIFs together with distal-end stress triaxiality

$$\sigma^* = \sigma_p + \frac{\rho_o}{\varepsilon_d} v_o^2 \quad (8)$$

In above equation,  $\sigma_p$  is the plateau stress,  $v_o$  is the initial velocity,  $\rho_o$  is the initial density and  $\varepsilon_d$  is the densification strain. The plateau stress of present AAC sample is determined using Eqn. 3 for the strain rate sensitive case (Model 2) as

$$\sigma_p = \sigma_s \left( \frac{v}{v_s} \right)^{1.026\alpha} \quad (9)$$

where,  $v$  is the impact velocity and  $v_s$  is the reference velocity corresponding to the reference strain rate. The determined experimental values of  $\sigma_s$  and  $v_s$  are 4.225 MPa and  $7.8 \times 10^{-7} \text{ m s}^{-1}$ , respectively. A constant plateau stress of 5.11 MPa is taken for the strain rate insensitive case (Model 1). The densification strain was determined from confining compression tests on AAC samples. Simply, a cylindrical sample was tightly fitted inside a cylindrical tube, then the sample was compressed with a flat end punch. Figure 11(a) shows the confinement stress-strain curves of these tests together with the picture of tested sample before and after the test. The densification of the sample is clearly seen in these pictures, showing a nearly full confined state. The densification strain is determined 0.28 by using a liner intercept method as shown in Figure 11(a).

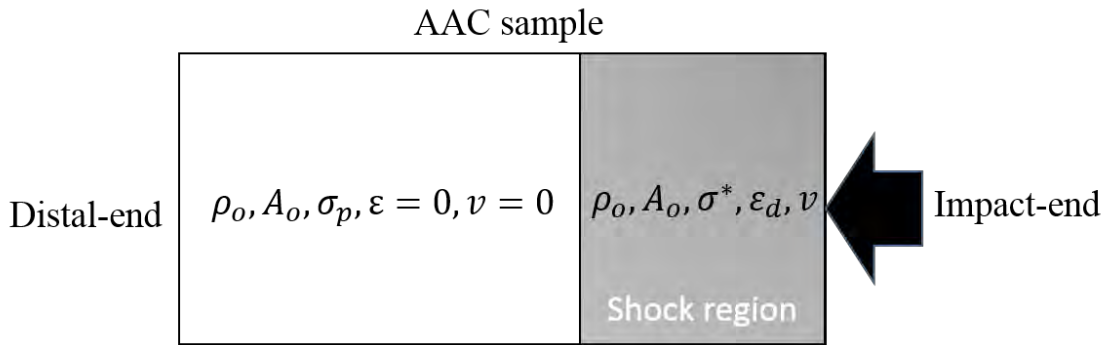


Fig. 11. Schematic of shock formation in AAC sample

The predicted impact-end stresses using Eqn. 8 and Eqn. 9 are shown as function of the logarithm of velocity in Figure 11(b) and (c) together with experimental and numerical distal-end and impact end DIFs, respectively. The predictions result in critical velocities near  $10 \text{ m s}^{-1}$  for the increased impact-end stresses for 1D state of strain as shown in the same figures. This also shows a good match with the experimental and numerical results.

**Conclusions.** The effect of inertia and strain rate on the failure of a cellular concrete ( $600 \text{ kg m}^{-3}$ ) was investigated using a constant volumetric-failure-strain erosion criterion in LSDYNA. Two modelling approach, namely rate insensitive (constant compressive yield stress) and rate sensitive (variable compressive yield stress) were implemented and the results were compared with the experimental compression tests performed at similar strain rates, between  $2 \times 10^{-3} \text{ s}^{-1}$  and  $\sim 4150 \text{ s}^{-1}$ . The effect of inertia in increasing the compressive strength of cellular concrete at increasing strain rates was shown both experimentally and numerically. An “s-type” compressive strength relation with strain rate was also found, composing of three different distinct regions: a lower velocity-dependent strength region at the quasi-static velocities (Region 1), a higher velocity-dependent strength region at intermediate velocities (Region 2) and again a lower velocity-dependent strength region above about  $1150 \text{ s}^{-1}$  (Region 3). In experimentally tested samples, a shock compressive strength was presumed to be reached in Region 2 or just at the beginning of Region 3, resulting in a cut-off DIF value (2.78), while in numerically tested samples, the compressive strength (Model 2) increased even in Region 3 with a rate very much similar to that of Region 1. One dimensional state of strain condition above a critical velocity was also shown numerically and the stress triaxiality increased to 0.66 within 1 and  $30 \text{ m s}^{-1}$ , reaching a fully constraint 1D state of strain condition above  $30 \text{ m s}^{-1}$ . In accord with this, the numerical fracture mode, as with the experiments, switched from an axial-dominated to a radial-dominated cracking after about  $20 \text{ m s}^{-1}$ . A simple shock analysis also proved 1D state of strain after about  $10 \text{ m s}^{-1}$ . Finally, the strain rate dependent compressive strength was numerically presented as partly due to the change of the deformation state from a 1D state of stress to an 1D state of strain and partly due to the intrinsic rate sensitivity of cellular concrete.

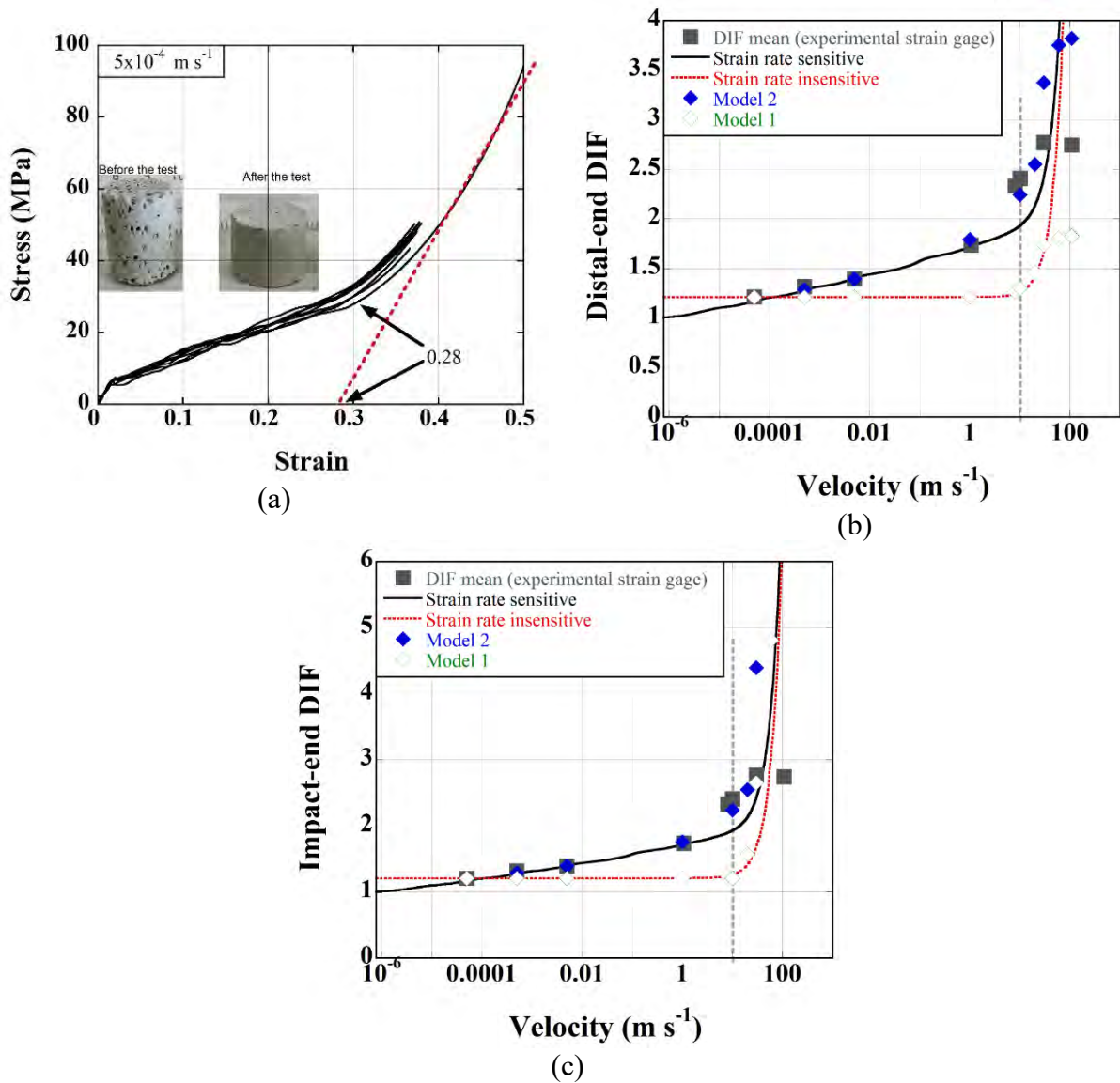


Fig. 12. (a) confined compression stress-strain curves and (b) distal-end and (c) impact-end stress prediction based on  $r$ - $p$ - $p$ - $l$  model

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